#### Mechanical Degradation of Ferritic/Martensitic and Austenitic Steels in CO<sub>2</sub> Environment



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# National Energy Technology Laboratory



#### **Research and Innovation Center**

- The Advanced Turbines Program at NETL conducts R&D for directly and indirectly heated supercritical carbon dioxide-based power cycles for fossil fuel applications.
- The focus is on components for indirectly heated fossil fuel power cycles with turbine inlet temperature in the range of 1300 -1400°F (700 - 760°C) and oxy-fuel combustion for directly heated supercritical CO<sub>2</sub> based power cycles.
- Materials issues involve quantification of creep, fatigue, oxidation and other mechanical and chemical processes in sCO<sub>2</sub> and gaseous CO<sub>2</sub> environments.









https://netl.doe.gov/onsite-research/materials

https://netl.doe.gov/node/7552



# Introduction

#### Oxidation:

- 9% Wt. Cr steels under CO<sub>2</sub>
  - Form duplex Fe-rich oxide scales following parabolic growth
  - Subject to enhanced carburization
- Austenitic Stainless (18-25% Wt. Cr.)
  - Protective Cr-rich scale is known to be interrupted in CO<sub>2</sub>







#### <u>Mechanical Properties:</u>

- Exsitu tests not showing difference after 500+ hour exposures, except in Base metal thin sections or with additions of H<sub>2</sub>O, O<sub>2</sub> gases
  - Both steels and superalloys
- Insitu testing is more complex
  - Ethelene cracking experiments show severe carburization
  - Autoclave testing IN600 alloy only showed loss of creep life in sCO<sub>2</sub>







#### **Materials**



- P91, 9% Cr steel, Martensitic / Ferritic
- MARBN, 9% Cr steel, Martensitic
- 347H, 17.3% Cr stainless steel, Austenitic with Nb-rich carbides









### **Methods**



- Oxidation experiments performed in tube furnace
  - 2.54, 0.6 and 0.5 mm thick dogbone specimens
  - Surface finish of 600 grit
  - 650°C, CO<sub>2</sub> with 4%  $H_2O$  and 1%  $O_2$  by volume.
  - 10<sup>-6</sup> Torr Vacuum
  - 1000-hour exposure
- Tensile tests
  - 2.54, 0.6 and 0.5 mm thick dogbone specimens preexposed as above
  - ASTM E-8 test methodology
  - 3.3\*10<sup>-5</sup> / s strain rate to failure
- Creep testing
  - Cylindrical specimens
  - No pre-exposures
  - Refort tube controls environment at ambient pressures
    only
  - 207 MPa applied load at 650°C tested to failure





# Pre-Exposure / Oxidation Results



- Tensile specimens were exposed to DF4 environment for 1000 h prior to testing
- P91 showed similar mass gain for both thicknesses
- Spallation led to reduction in mass gain for thin 347H specimen
- Thicker 347H specimen only had ~15% of the mass gain of P91 under the same conditions

Alloy	Specimen Thickness (mm)	Mass gain (mg/cm²)
P91	0.50	29.0
P91	2.54	28.0
347H	0.60	-6.0
347H	2.54	4.6



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#### P91 Steel

- No significant changes noted from specimen thickness under you wacuum exposures
- Thin specimen failed before yielding
- Thicker specimen's YS and UTS unchanged, but elongation was reduced by 57%

#### <u>347H</u>

 Only notable change from the thin section, which YS doubled and elongation severely reduced





# Tensile Fracture Surfaces – P91

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- Thick specimen showed
  - ~0.8mm enhanced carburization zone around the perimeter
  - No shear lip formation, suggesting embrittlement
- Thinner specimen showed
  - Severe embrittlement
  - Electron backscatter shows oxidation/carburization effected zone penetrating to ~0.2 mm (including oxidation scale)





# Tensile Fracture Surfaces – 347H

- Only Thin specimen featured
- Specimen failed at 45° incline plane, suggesting shear ductility
- High magnification shows phase contrast ~50 µm deep
- Microprobe analysis verifies this to be oxidation related

S. DEPARTMENT OF



20.0



C %wt

- MARBN and 374H tested to date
- Tests conducted in controlled environment retort tube
  - 100 sccm CO2, 650°C, ~207 MPa
- MARBN alloy showed
  - ~3.5X reduction in time to failure
  - ~1 decade faster MCR
- Faster MCR suggest mechanisms change



Alloy Designation	Test Temperature (°C)	Stress Level (MPa)	Time to Failure (h)	Elongation to Failure (%)	Reduction in Area (%)	Minimum Creep Rate (%/h)	Larson Miller Parameter, C=25
MARBN - Air	650	206.84	527	13.2	64.4	2.21·10 <sup>-3</sup>	25586.18
MARBN - Air	650	206.84	545	12.6	69.9	1.74·10 <sup>-3</sup>	25605.15
MARBN - CO <sub>2</sub>	650	206.84	169	16.1	67.3	8.20·10 <sup>-3</sup>	25129.21
MARBN - CO <sub>2</sub>	650	206.84	140	14.7	58.3	1.00·10 <sup>-2</sup>	25054.18



## Fracture Surfaces - MARBN



- Left images are from air, right from gaseous CO<sub>2</sub>
- Brittle failure near specimen edge
- Cross sectioned samples
  - Hardness and etching revealed carburized zone penetrated ~200 µm deep





Rozman et al., 2021, Mat Sci Engr A, 826, #141996



### **Cross Sections - MARBN**



- Found complex interactions between oxide growth and enhanced carburization
- Carburization tied up Cr
- Less Cr to form protective chromia scale
- Feed back mechanism to enhance environmentally assisted cracking
- As evidenced by microprobe analysis and change in MCR







### Creep Results – 347H

- Creep tests on going
  - 650C, 197 MPa
  - Evaluating both effect of thickness and environment

10<sup>1</sup>

**10**<sup>0</sup>

**10**<sup>-1</sup>

**10**<sup>-2</sup>

**10**<sup>-3</sup>

Strain (mm/mm)

- To date no change NETL has found
  - Reduced creep life
  - Slight increase in MCR
  - Suggests no mechanism change





### Conclusions

#### 9%Wt Cr steels

- Carburization strongly effects mechanical properties
- Strategies to mitigate carburization are necessary if used in sCO<sub>2</sub> applications
- Thin sections subject to severe degradation
- Creep lifetimes reduced ~3.5X







### Conclusions

#### Austenitic steels – 17%Wt Cr

- Generally better resistance to environmental effects in sCO<sub>2</sub>
- Thin sections showed brittle behavior (reduction in ductility and increase in yield stress)
- Creep experiments are on going
- Preliminary results show reduction in lifetimes but no change in MCR
- Suggests they may be resistant to environmentally assisted cracking



**10**<sup>-1</sup>

100

Time (h)

10<sup>1</sup>

10<sup>2</sup>

 $10^{3}$ 

**10**<sup>-2</sup>



10<sup>3</sup>



 $10^{-2} \ 10^{-1} \ 10^{0} \ 10^{1} \ 10^{2}$ 

Time (h)

# Thank You

#### Questions?

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# NETL Resources

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